

Results of FITT Oxidation Study

D.J. Skitt, T.J. Gerczak, R.L. Seibert, J.W. Werdren, Z.M. Burns, G.W. Helmreich, J.D. Hunn

ART Program Review July 12th, 2022

Nuclear Fuel Development Section

Oak Ridge National Laboratory

ORNL is managed by UT-Battelle, LLC for the US Department of Energy



Outline

- Objectives/Background
- Experiment design
- Particle failure fraction analysis
 - Increase in irradiated particle failure response compared to that of unirradiated
 - External factors most likely caused more failures than anticipated
- Oxide formation kinetics
 - Similar oxide growth rate between irradiated and unirradiated particles
- Microstructure examination
 - Crystalline oxide structures observed in all samples
- FITT Update

Objectives

- Measure TRISO particle failure fraction as a function of oxidizing conditions
 - Provide data to support Air/Moisture-Ingress Experiment (AMIX) test plan development
- Compare oxidation behavior of the SiC layer in unirradiated and irradiated TRISO particles
 - Analyze differences in oxidation kinetics between unirradiated and irradiated SiC
 - Observe oxidation microstructure in individual TRISO particles



FITT: Furnace for Irradiated TRISO Testing

- Flexible, intentionally-simple, costeffective capability to heat small batches of irradiated TRISO particles up to 1700°C over times >1500 h outside a hot cell
 - Closed-bottom ceramic tube in box furnace containing 10–30 particles under flowing inert gas or oxidizing environments (up to 21% O₂)
 - Installed in the Irradiated Fuels Examination Laboratory (IFEL) radiological facility at ORNL where AGR hot cell work is performed
 - Intended to support integral release/oxidation tests in the Core Conduction Cooldown Test Facility (CCCTF), Fuel Accident Condition Simulator (FACS), and AMIX systems



Image of FITT system in IFEL hood



TRISO Particle Sample Selection

- Irradiated particles from AGR-2 Compact 5-4-2 had outer pyrolytic carbon (OPyC) layer "burned back" to expose SiC
 - Exposed SiC layer to initiate oxidation growth at t = 0
 - Not expected to influence failure fraction based on previous post-burn failure fraction measurements [1]
 - Resulting oxide thickness extrapolated to be <0.01 µm [2]
- Unirradiated particles from Compact LEU09-OP2-Z002
 - Subjected to a similar burn back process as Compact 5-4-2



Time-averaged, volume-averaged (TAVA) temperature versus burnup for AGR-1 and AGR-2 UCO Compacts [3]

[1] Hunn, J.D., F.C. Montgomery, and P.J. Pappano. 2010. Data Compilation for AGR-2 UCO Variant Compact Lot LEU09-OP2-Z, ORNL/TM-2010/017, Revision 1. Oak Ridge: Oak Ridge National Laboratory.

[2] Cao, Fangcheng, D, Zhang, Q. Chen, H. Li, and H. Wang. 2020. "Evaluation of Oxidation Performance of TRISO Fuel Particles for Postulated Air-Ingress Accident of HTGR." Journal of Chemistry 2020, Article ID 6568987: 8 pages. doi: 10.1155/2020/6568987.

ional Laboratory [3] Stempien, John D, J.D. Hunn, R.N. Morris, T.J. Gerczak, and P.A. Demkowicz. 2021. "AGR-2 TRISO Fuel Post-Irradiation Examination Final Report," INL/EXT-21-64279-Rev000. Idaho Falls: Idaho National Laboratory

Experimental Approach

- Ten irradiated particles and ten unirradiated particles loaded into a SiC cup separated by a partition
- Once at temperature, the flow gas was switched from Argon to oxidant.
 - Dispersed ~1" above the SiC cup at 50 mL/min (internal volume of system was ~321 mL)
 - Refreshes system atmosphere every ~6 min
- Sample location temperature confirmed with separate thermal profiling test
- Hach Orbisphere K-M1100 Oxygen Sensor
 - Measured oxygen concentration from system exhaust
 - Confirmed no oxygen starvation during 2% O₂ exposure

[1] Presser and Nickel, "Silica on Silicon Carbide", Critical Reviews in Solid State and Materials Sciences, 33:1, 2008, 1–99.

Conditions of six oxidation tests performed

Temperature (°C)	Atmosphere	Exposure Times (h)
1200	$21\% O_2$ (balance N ₂)	400
1400	$21\% O_2$ (balance N ₂)	50, 100, 200, 400
1400	2% O ₂ (balance He)	400



CAK RIDGE National Laboratory

Irradiated TRISO particle failure fraction increased with exposure time and temperature



 Failed SiC layer would be indicated by ¹³⁷Cs release

CAK RIDGE

No low ¹³⁷Cs particles detected by IMGA

Particle failure during various oxidation tests



Exposure

- Complete failure resulted in particle consumption by system and lack of recovery
- Evidence of particle fragments/debris at bottom of SiC cup

External factors from experimental approach influencing particle failure fraction

- No trend observed in failure rate of unirradiated particles
 - Suggested variables besides oxidizing conditions promoted particle failure
 - Previously observed in inert heating tests of burned back Compact 5-4-2 particles [1]
- Possible factors biasing particle failure
 - Nitric acid exposure during burn back process
 - SiC cup interaction with particles
 - SiC shell damage from handling
 - Variations in oxygen flow rate
 - Temperature gradients

Particle failure during various oxidation tests



CAK RIDGE

[1] Gerczak, T.J., Z.M. Burns, D.J. Skitt, R.N. Morris, and J.D. Hunn, 2020. AGR 2 loose particle heating tests in the furnace for irradiated TRISO testing, ORNL/TM-2020/1715-R0. Revision 0. Oak Ridge: Oak Ridge National Laboratory.

Irradiated particle failure fraction increased with longer oxidation exposure times

- Failure fraction increased from 200 h to 400 h of exposure, as well as from 1200°C to 1400°C
- Higher failure rate observed in FITT than historic KORA tests at 1400 °C [1]
 - KORA experienced first TRISO failure (10% of particles) after 397 h of exposure to air





[1] International Atomic Energy Agency. 1997. Fuel performance and fission product behaviour in gas cooled reactors. IAEA-TECDOC-978. Vienna: International Atomic Energy Agency.

Oxide Thickness Analysis



- Increased oxidation response for prolonged exposures at 1400°C, 21% O₂
- Decreasing pO_2 only reduced average oxidation rate at 1400°C by 5–15%
 - Supported by prior studies for low pO₂ oxidation of SiC at similar exposures [1,2]



[1] Narushima, T., T. Goto, T. Hirai, and Y. Iguchi. 1997. "High-Temperature Oxidation of Silicon Carbide and Silicon Nitride." CAK RIDGE Materials Transactions, JIM 38 (10): 821-835. National Laboratory [2] Goto, T., Homma, H. 2002. "High-temperature active/passive oxidation and bubble formation of CVS SiC in O2 and CO2

atmospheres." J. of European Ceramics Society, 22 (2002) 2749-2756.

Effect of pO_2 on SiC oxidation rate [2]

1400°C Oxidation Kinetics



- Deal-Grove model gives non-physical fit for long exposure times (x = B(t/x) - A)
- Parabolic oxidation rate used instead (x² = Bt)
 - Previous parabolic rate studies typically observed at < 48h timescale [1,2]
- Longer (400 h) exposures resulted in deviation from parabolic oxide growth



[1] Cao, Fangcheng, D, Zhang, Q. Chen, H. Li, and H. Wang. 2020. "Evaluation of Oxidation Performance of TRISO Fuel Particles for Postulated Air-Ingress Accident of HTGR." Journal of Chemistry 2020, Article ID 6568987: 8 pages. doi: 10.1155/2020/6568987.

[2] Liu, Rongzheng, B. Liu, K. Zhang, M. Liu, Y. Shao, C. Tang, 2014. "High temperature oxidation behavior of SiC coating in TRISO coated particles." Journal of Nuclear Materials 453, 107-114.

Parabolic rate constant, B, for various exposure ranges

	Unirradiated		Irradiated	
Exposure	Β (μm²/h)	R ²	B (μm²/h)	R ²
50–100 h, 1400°C	0.055	0.998	0.039	0.963
50–200 h, 1400°C	0.062	0.996	0.055	0.970
50–400 h, 1400°C	0.120	0.928	0.143	0.889
0–48 h, 1300°C (Literature) [1,2]	0.042-0.058			
0–48 h, 1400°C (Literature) [1,2]	0.121-0.126			
400 h, 1200°C	0.018		0.015	
0-48 h, 1200°C (Literature) [1,2]	0.017–0.036			

- Deviation from parabolic growth observed at 1400°C, 400 h
 - Attributed decomposition and complexity of oxide structures at longer exposure times [3]
- Rate constant for 1400°C, 50-200 h range similar to literature on unirradiated TRISO oxidation at 1300°C
- Single observation for rate constant at 1200°C in agreement with literature

[1] Cao, Fangcheng, D, Zhang, Q. Chen, H. Li, and H. Wang. 2020. "Evaluation of Oxidation Performance of TRISO Fuel Particles for Postulated Air-Ingress Accident of HTGR." Journal of Chemistry 2020, Article ID 6568987: 8 pages. doi: 10.1155/2020/6568987.

CAK RIDGE

[2] Liu, Rongzheng, B. Liu, K. Zhang, M. Liu, Y. Shao, C. Tang, 2014. "High temperature oxidation behavior of SiC coating in TRISO coated particles." Journal of Nuclear Materials 453, 107-114.
 [3] Costello, J.A., and R.E. Tressler. 1986. "Oxidation Kinetics of Silicon Carbide Crystals and Ceramics in Dry Oxygen." Journal of the American Ceramics Society 69 [9]: 674–681.

Surface Oxide Structural Differences at 1400°C

• 50–100 h

- Unirradiated oxide surfaces showed coarser microstructures than that of the irradiated particles
- 200–400 h
 - Similar topography between irradiated and unirradiated samples





Unexposed Compact 5-4-2 particle SiC layer

Unirradiated Sample



Irradiated Sample



Irradiated Sample

5 µm

Unirradiated Sample

1400°C, 50 h

1400°C, 200 h

400 h surface layer observations varying temperature and O₂ concentration



1200°C, 21% O₂

1400°C, 21% O₂

1400°C, 2% O₂

 1200°C, 400 h oxide topography differences between particle types align with observations from 50 h and 100 h exposures at 1400°C



Non-uniform secondary oxide

- Certain particles had a secondary oxide deposited on top of the primary oxide layer
 - Non-uniform and localized, compared to uniform primary oxide
 - Theorized to be the result of oxide buildup at the contact points between the particles and SiC cup
- Focused liftouts on regions away from contact points during primary oxide thickness analysis



Example of secondary oxide formed on a 1400°C, 100 h particle



Oxide Microstructure Characterization

- SiC/oxide lamellas prepared with FIB/SEM
- S/TEM analysis
 - General characterization to confirm microstructure
- EDS performed to examine fission product presence



Unirradiated Irradiated

1400°C, 100 h

EDS mapping of the (A) 50 h irradiated sample and (B) an unirradiated sample, which show element locations within the chosen interface areas [1].

- Previous studies suggest an increased oxidation rate due to appreciable impurities present at the SiC/oxide interface [2].
- Note: Pd signal intensity was increased to observe Pd features in the SiC layer. Noise in the oxide layer does not suggest Pd presence.

[1] Skitt, D.J., R.L. Seibert, T.J. Gerczak, J.D. Hunn, Z.M. Burns, G.W. Helmreich. 2021. Oxidation Testing and Examination of AGR-2 TRISO Particles, ORNL/TM-2021/2092, Revision 0. Oak Ridge: Oak Ridge National Laboratory

[2] Singhal, S.C., and F.F. Lange. 1975. "Effect of Alumina Content on the Oxidation of Hot-Pressed Silicon Carbide." *Journal of the American Ceramic Society* 58 (9-10): 433–435.

All oxide microstructures examined were crystalline

- Irradiated and unirradiated samples revealed similar crystalline microstructures
 - Could not analyze differences due to amorphization from 200 kV electron beam
- 1400°C, >200 h irradiated particles exhibited larger grain size than that of unirradiated samples



1200°C, 400 h Unirradiated OAK RIDGE National Laboratory

1400°C, 400 h Unirradiated



1400°C, 200 h Unirradiated

1400°C, 200 h Irradiated

1400°C, 50–200 h oxidation rate was slightly slower in irradiated particles, but faster after 400 h

	Unirradiated		Irradiated	
Exposure	B (μm²/h)	R ²	B (μm²/h)	R ²
50–100 h, 1400°C	0.055	0.998	0.039	0.963
50–200 h, 1400°C	0.062	0.996	0.055	0.970
50–400 h, 1400°C	0.120	0.928	0.143	0.889
0–48 h, 1300°C (Literature) [1,2]	0.042–0.058			
0–48 h, 1400°C (Literature) [1,2]	0.121-0.126			
400 h, 1200°C	0.018		0.015	
0-48 h, 1200°C (Literature) [1,2]	0.017–0.036			

- Irradiated and unirradiated oxidation kinetics overall were similar
- Increase in irradiated oxidation rate at 400 h was theorized to be the result of larger grain sizes
 - Possible contribution from presence of excess impurities (e.g., fission products) [1]
 - Oxidant diffusion previously noted to be faster in amorphous, quartz, and cristobalite structures [2]

CAK RIDGE
[1] Opila, E. 1995. "Influence of Alumina Reaction Tube Impurities on the Oxidation of Chemically-Vapor-Deposited Silicon Carbide." Journal of the American Ceramic Society 78 (4): 1107–1110.
[2] Costello, J.A., and R.E. Tressler. 1986. "Oxidation Kinetics of Silicon Carbide Crystals and Ceramics in Dry Oxygen." Journal of the American Ceramics Society 69 [9]: 674–681.

Summary

- Irradiated TRISO particles failure fraction increased with prolonged exposures at 1400°C.
 - Similar trend observed in KORA \rightarrow temperature and duration affects failure fraction
 - External factors contributed to uncorrelated particle failures
 - E.g., burnback particles, handling methods
- Irradiated oxidation rate was slower for 50–200 h but faster after 400 h at 1400°C, 21% $\rm O_2$
 - Difference between irradiated and unirradiated not substantial
- Minor variations observed suggest oxidation rates in unirradiated TRISO studies can be used to estimate irradiated TRISO oxidation kinetics.



Current and Future FITT Efforts

- AGR-5/6/7 inert gas thermal exposures in FITT
 - Examine ^{110m}Ag and ¹⁵⁴Eu release
 - Compact 2-2-1 particles (non-burnback)
 - Six 100 h tests, 1100-1600°C
- AGR-5/6/7 oxidation testing in 21% O_2 atmosphere
 - Compact 2-2-1 particles (non-burnback)
 - Two 1400°C exposures, 100 & 200 h
 - ZrO₂ cup to replace SiC cup to reduce interaction with particles





Questions? Darren Skitt skittdj@ornl.gov Hot cell activities were supported by Oak Ridge National Laboratory (ORNL) Irradiated Fuels Examination Laboratory staff. The authors would like to acknowledge Andrew Kercher and Dr. MEET. EXPLORE. LEARN Fred Montgomery for support with particle supply.

ORNL is managed by UT-Battelle, LLC for the US Department of Energy

